

# Diagnosing Adhesion and Cohesion for BEOL Reliability: From Die Seals and Crack Stops to Cu and Low k

## Low k Research

Max Gage, Markus Ong  
Taek-Soo Kim, Nathan Stein

## Chip/Package Interactions

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## Front End Structures

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Collaborators at INTEL, AMD, IBM, Applied Materials, ASM and JSR.

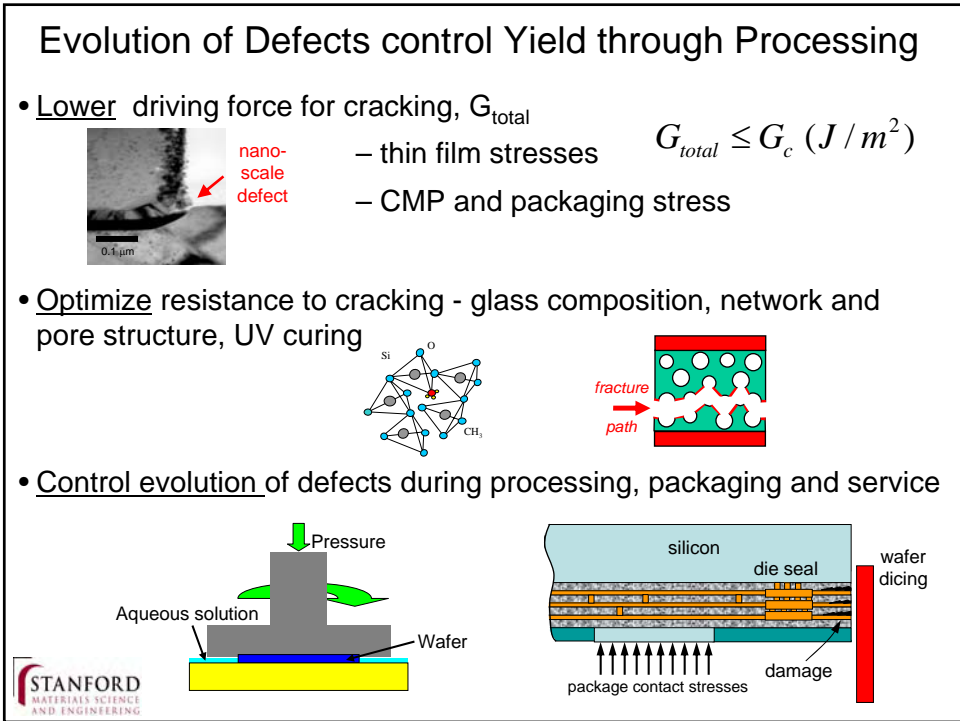
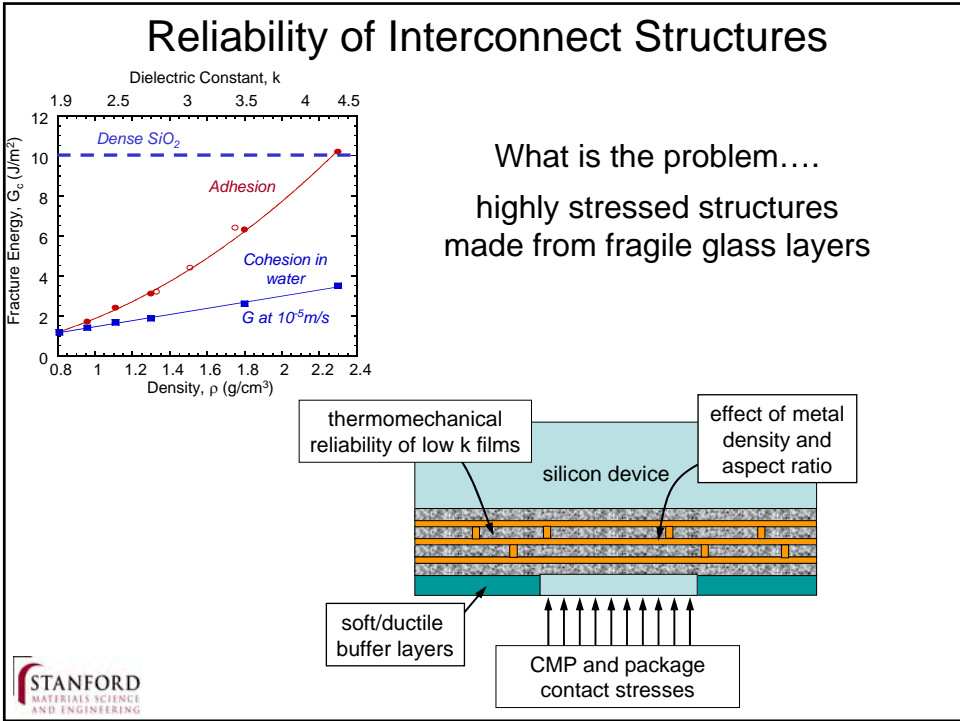


Work supported by the SRC, DOE, Device, Equipment and Materials Companies.

## Outline

- Mechanical Reliability of Low-k Materials
  - what is the problem?
  - fracture properties and scaling with density/dielectric constant
- UV Curing Effects on Mechanical Properties
  - fracture path and effects on cohesive and adhesive properties
  - balancing film stress, modulus and fracture properties
- Non-Uniform UV Curing Phenomena
  - evidence for non-uniform curing
  - likely mechanism associated with UV light interference
- CMP Aqueous Chemistry Effects
  - accelerated cracking in aqueous CMP slurries and cleaning solutions
- Die Seal and Crack Stop Structures
  - fracture mechanics approach to preventing dicing damage
- Summary





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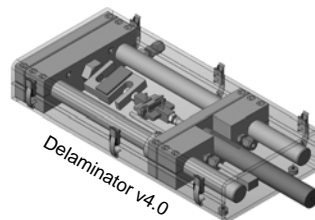
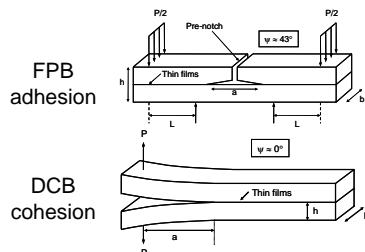
## Experimental Methods

### Glass Structure

- $^{29}\text{Si}$  and  $^{13}\text{C}$  magic angle spinning NMR (3.2 mm MAS rotor, 12-13 KHz spinning rates, 9.4 Tesla magnet)
- FTIR

### Fracture Mechanics Testing

- Four Point Bend (FPB) and Double Cantilever Beam (DCB) geometries



### Fracture Surface Morphology/Composition

- AFM, XPS



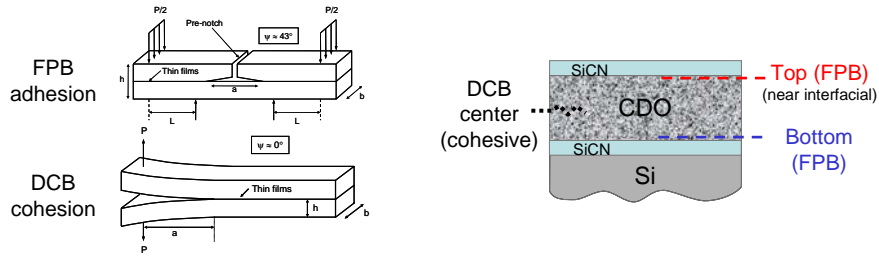
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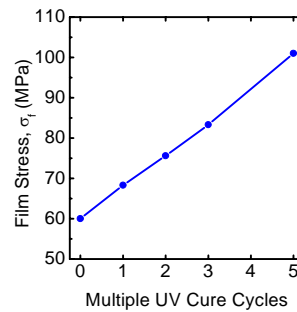
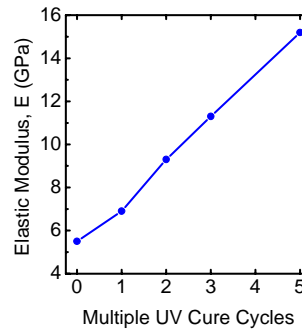
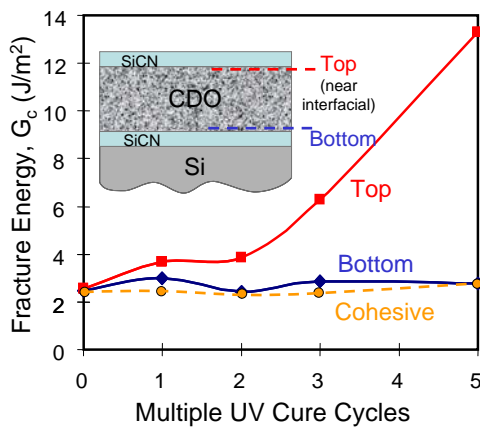


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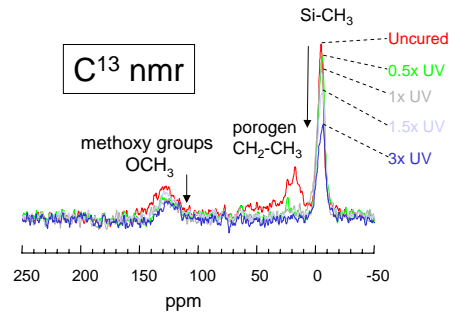
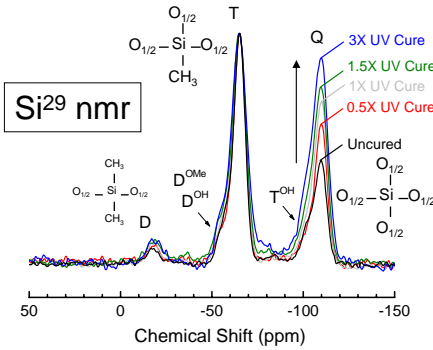
# Role of UV Cure on Mechanical Properties



Treatment	Dielectric Constant	Elastic Modulus, $E_f$ (GPa)	Residual Film Stress, $\sigma_f$ (MPa)	Crack Driving Force, $G$ ( $\text{J/m}^2$ )	Cohesive Fracture, $G_c$ ( $\text{J/m}^2$ )
none	2.62	5.5	60	0.287	2.50
UVx1	2.59	6.9	68.3	0.296	2.40
UVx2	2.62	9.3	75.6	0.269	2.36
UVx3	2.66	11.3	83.3	0.269	2.34
UVx5	2.72	15.2	101	0.294	2.66

# UV Curing Effects on Glass Structure

Porous DEMS ( $k = 2.5$ )



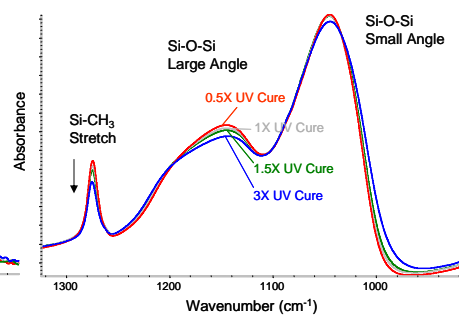
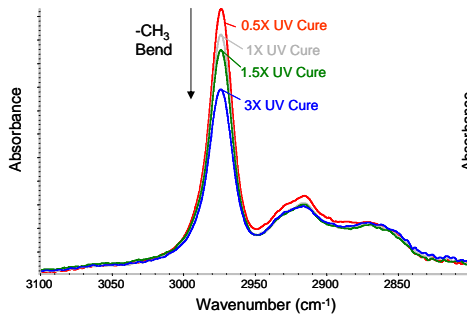
- progressive loss of  $-CH_3$  and  $-OH$  groups
- Increasing more Si-O-Si bridging bonds

- initial loss of  $CH_2$  groups from porogen
- progressive loss of Si- $CH_3$  groups
- some loss in C=C groups



Gage, Cui, Al-Bayati, Demos, MacWilliams, M'saad and Dauskardt – 2006.

# FTIR Structural Characterization



- Progressive reduction of  $-CH_3$  with UV curing
- Only minor change in skeleton (reduction in large angle Si-O-Si stretch band)
- No indication of transition to small angle network configuration with cure (insensitivity of the Si-O stretch bands at 1140  $cm^{-1}$  and 1040  $cm^{-1}$ )
- Improvements in fracture energies not due to rearrangement of the glass backbone but to an increased degree of network connectivity



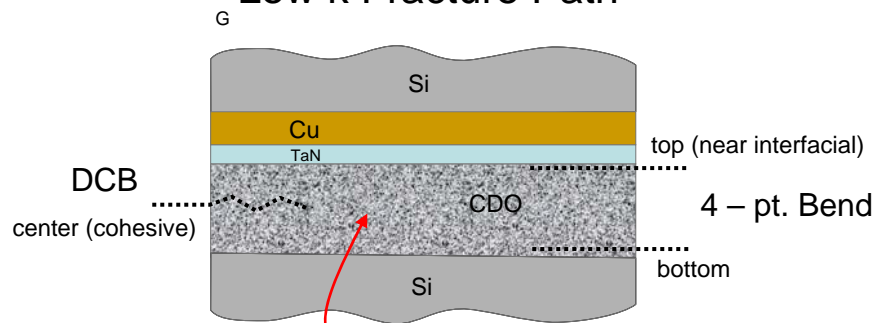
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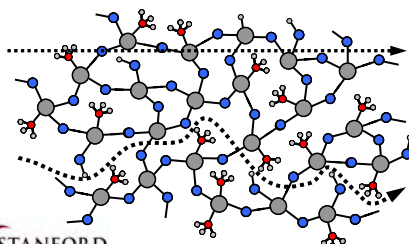
- Mechanical Reliability of Low-k Materials
  - what is the problem?
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- Studies of Glass Network Structure
  - NMR and FTIR reveal glass network structure
  - UV cure reactions and changes in network structure
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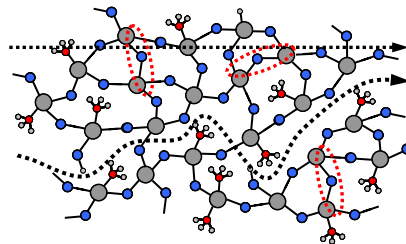
## Low k Fracture Path

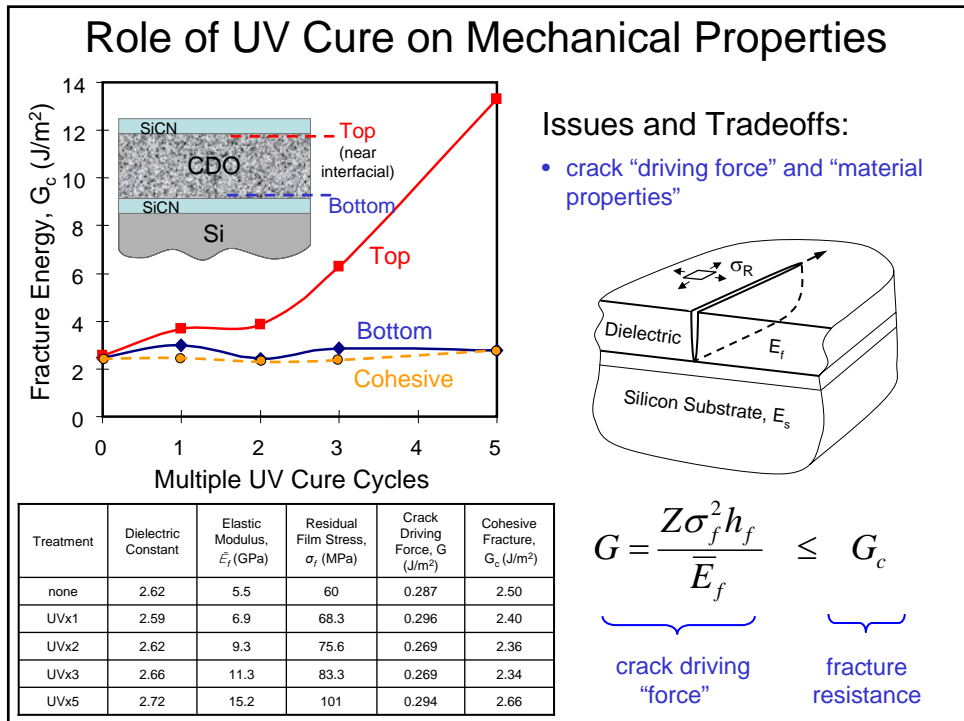
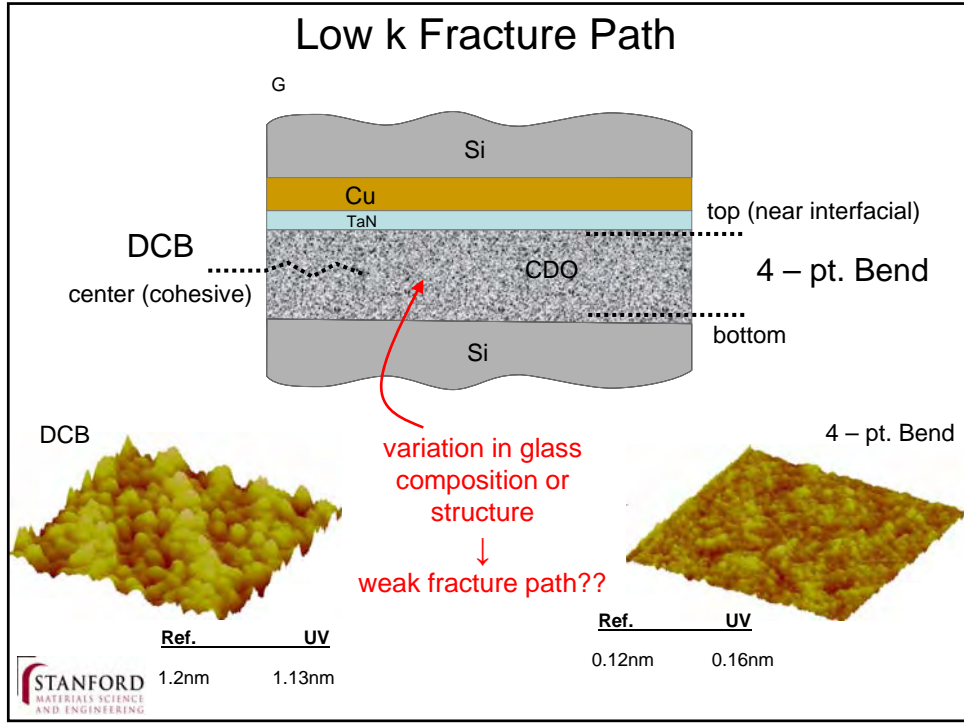


Uncured

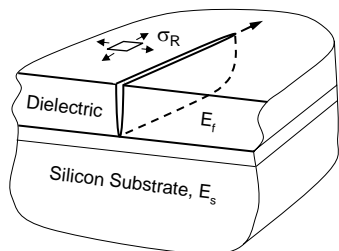


Cured





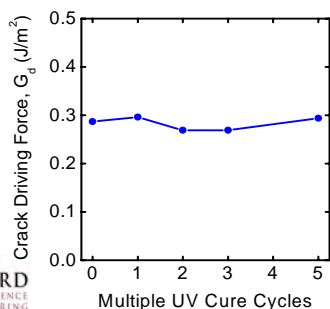
## Issues and Tradeoffs



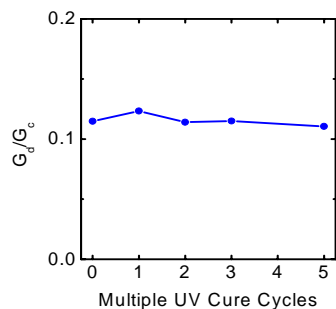
$$G_d = \frac{Z\sigma_f^2 h_f}{\bar{E}_f} \leq G_c$$

crack driving "force"
fracture resistance

$G_d$  vs. UV Cure Cycles



$G_d/G_c$  vs. UV Cure Cycles



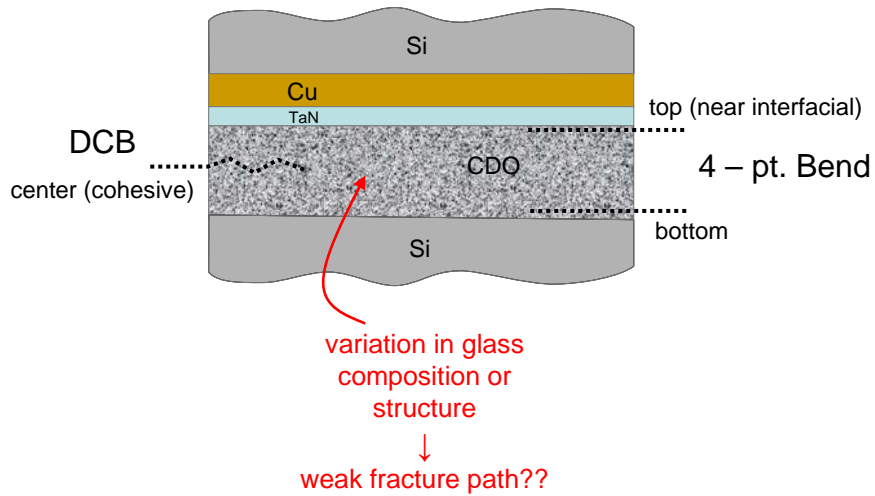
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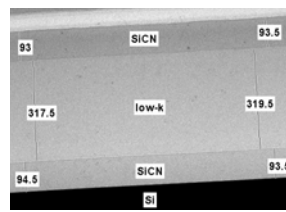
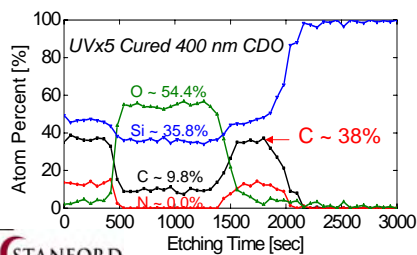
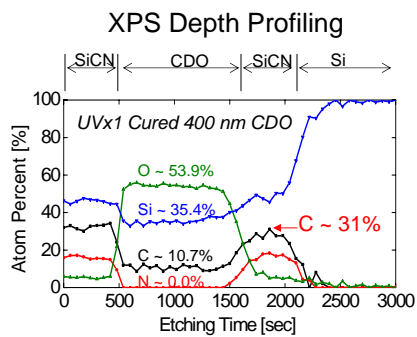




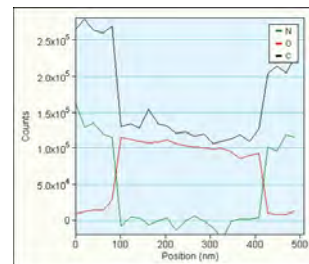
# Non-Uniform Curing Phenomena



# No Evidence of Compositional Variation

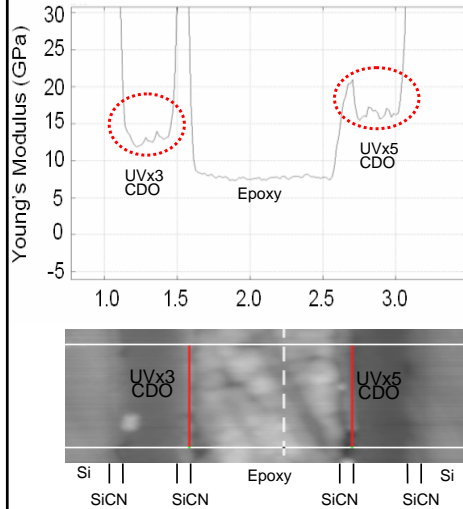


EELS composition profile taken in STEM mode

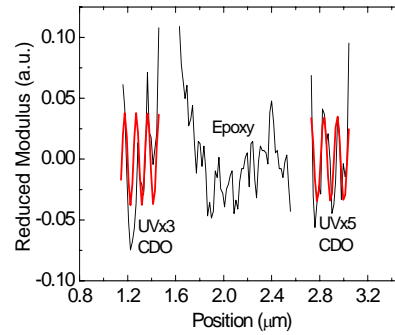


T. Kim, N. Tsuji, N. Kemeling, K. Matsushita, D. Chumakov, E. Zschech, and R. H. Dauskardt, AMC2006.

## FM-AFM Measurements of Modulus Variations

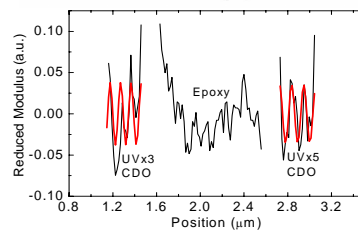
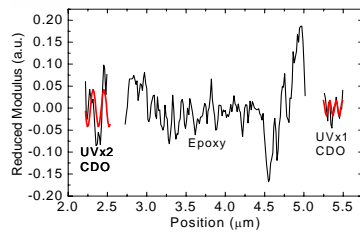
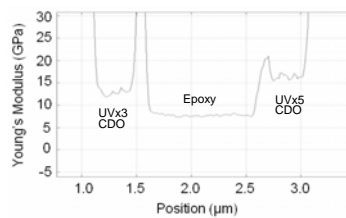
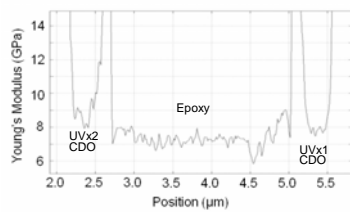
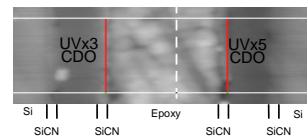
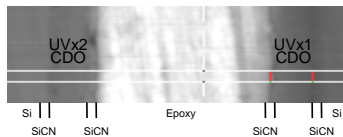


*sine wave fitting of  
modulus variations*



T. Kim, N. Tsuji, N. Kemeling, K. Matsushita, D. Chumakov, E. Zschech, and R. H. Dauskardt, AMC2006.

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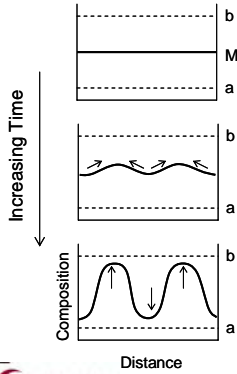


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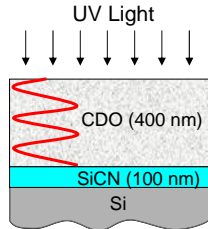
# Possible Causes of Modulus Variation

## Phase separation

Spinodal decomposition  
[Cahn and Hilliard, 1958]



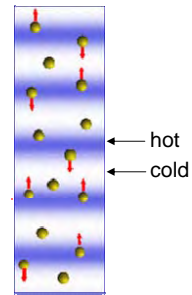
## UV light interference



Cured bond density proportional to light intensity

## Soret effect

Photothermal heat generated by UV standing wave and activated atoms migrate.



# Standing-Wave Effects during UV Curing

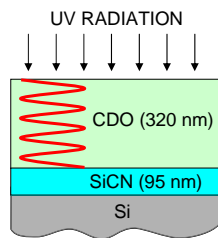
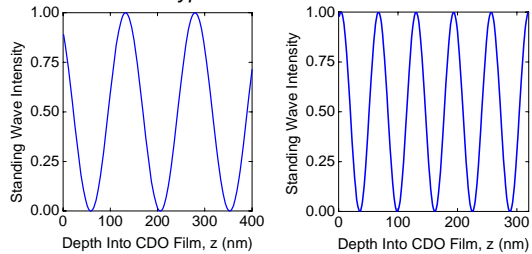
Simple Model: standing wave in a thin film on a perfectly reflecting substrate:

$$I = \alpha \sin^2 k(d - z) \quad \text{with } k = 2\pi n / \lambda$$

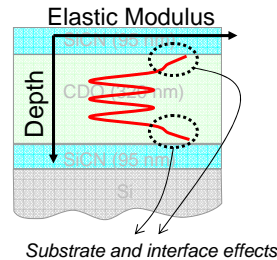
where  $\alpha$  amplitude of the standing-wave  
 $n$  film refractive index  
 $\lambda$  wavelength of the incident light  
 $d$  film thickness  
 $z$  depth into film

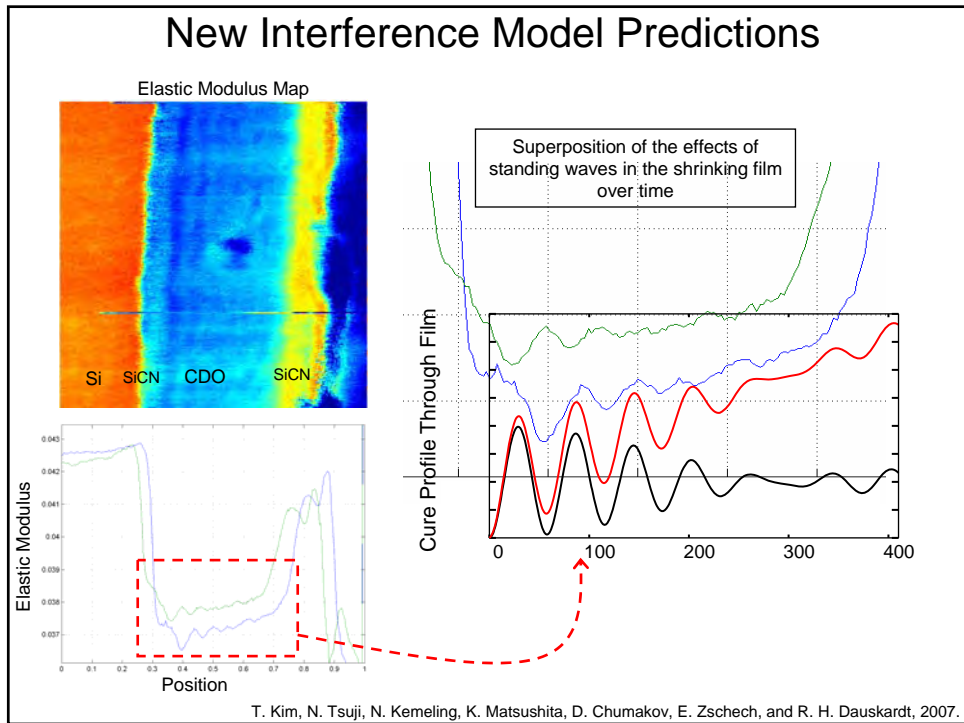
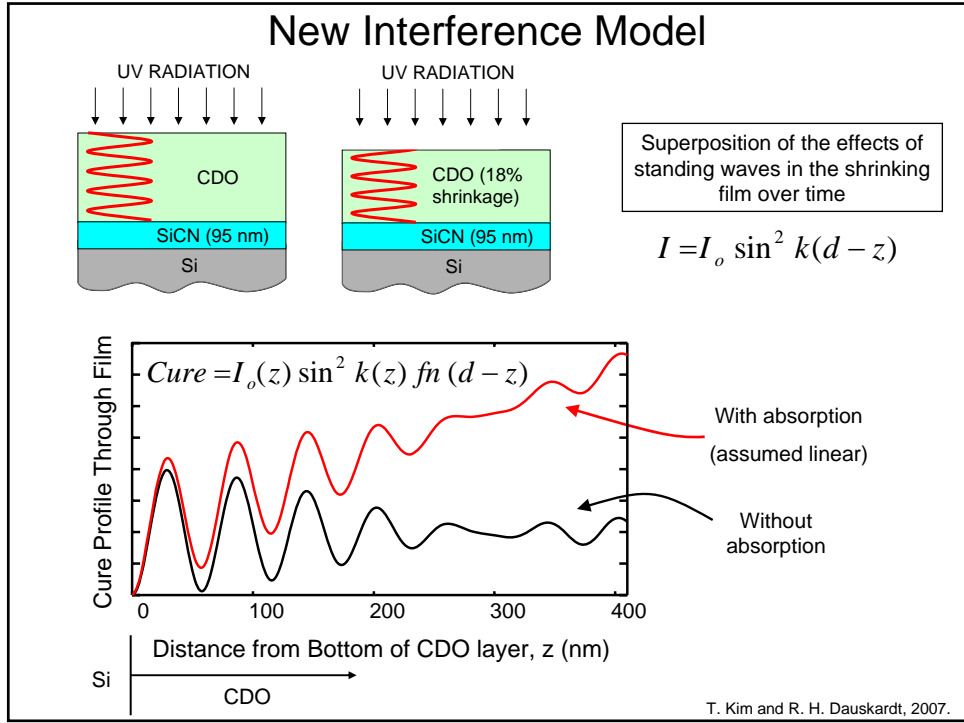
Cuthbert, J. D. *Optical projection printing*.  
 Solid State Technology 50, 59-69 (1977).

Typical values of  $n$  and  $\lambda$

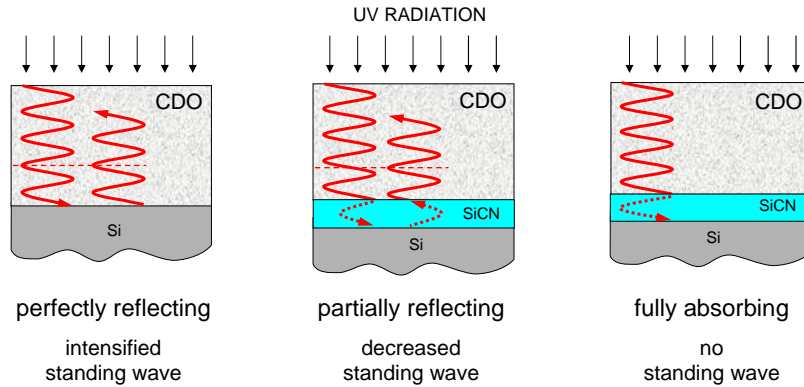


Measurement of modulus variation by FM-AFM





## Role of the Bottom SiCN Layer



- Reflectivity of bottom interface strongly dependent on underlying layer.
- SiCN thin film can significantly absorb high energy UV light depending on its carbon content

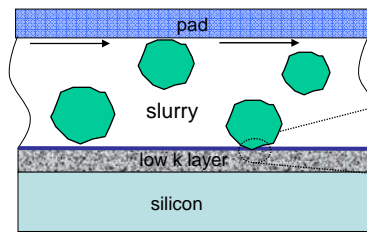


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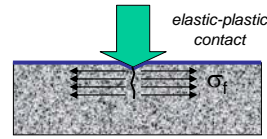
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# Mechanics of Damage Initiation and Propagation

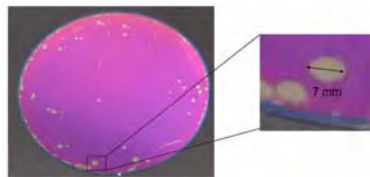


Mode I  
crack initiation and propagation



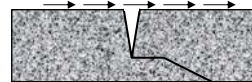
damage created by slurry particle or pad asperity

damage involving low-k film delamination



tractions on film edge of a through cut induce a Mode III component responsible for delamination shape

Mode II crack deflection



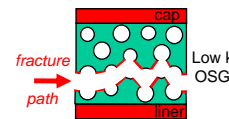
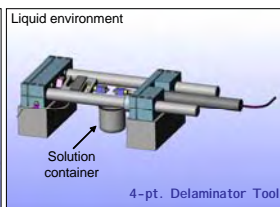
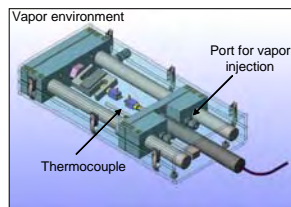
crack kinks to interface for  $K_{II} > 0$



crack kinks to surface for  $K_{II} < 0$

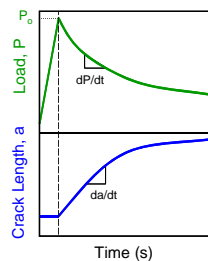


# Automated Crack Velocity Testing

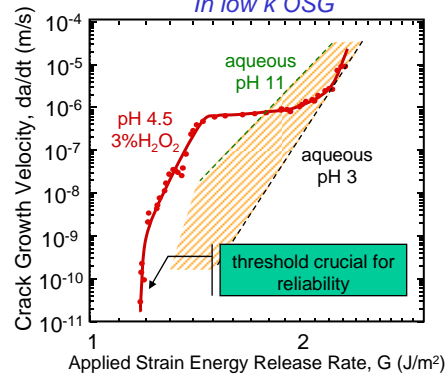


Accelerated Cracking  
In low k OSG

Load Relaxation Crack Growth Technique

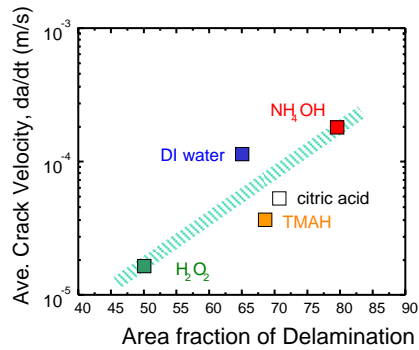
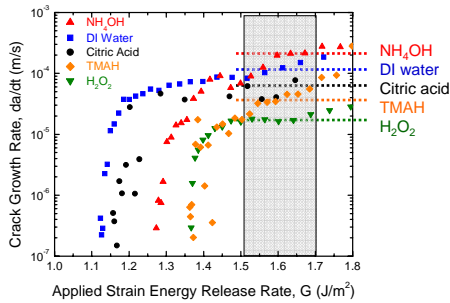
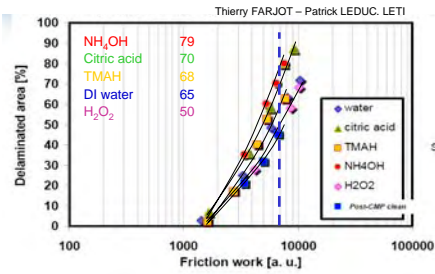
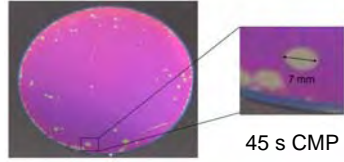


auto analysis



System and support:  
DTS Company, Menlo Park, CA ([dauskardt@stanford.edu](mailto:dauskardt@stanford.edu))

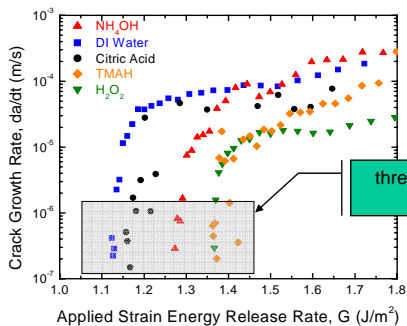
# Extensive Damage during CMP



# Reduced Damage and High Yield in CMP

- low crack growth rates critical for growth of nano-scale flaws
- dominated by threshold behavior in v-G curves

Crack growth rates  $<10^{-10}$  m/s (below threshold) necessary to achieve reliable integration



Synergistic effects of CMP slurry chemistry and stress on defect evolution/crack growth are unknown!

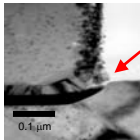
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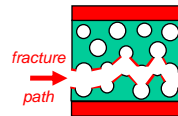
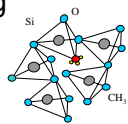
## Evolution of Defects control Yield through Processing

- Lower driving force for cracking,  $G_{total}$ 
  - thin film stresses
  - CMP and packaging stress
$$G_{total} \leq G_c \text{ (J/m}^2\text{)}$$

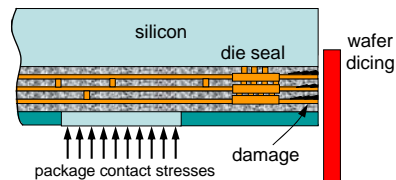
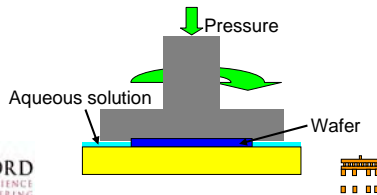


nano-scale defect

- Optimize resistance to cracking - glass composition, network and pore structure, UV curing



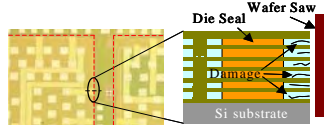
- Control evolution of defects during processing, packaging and service



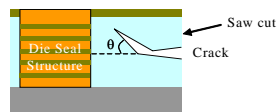
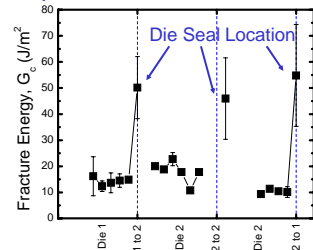


## SRC 1391.001: Materials and Interface Innovation for New Concepts in Microelectronic Packaging

Die Seal structures to protect interconnect from saw damage

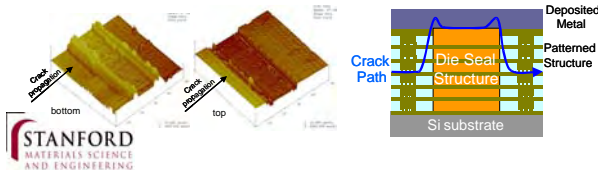


Improved Fracture Resistance

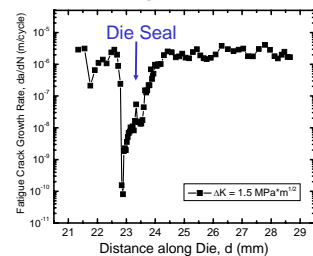


Effectiveness of die seal depends on its ability to deflect cracks

AFM images of paired fracture surfaces show large deflections around die seals



Reduced Fatigue Crack Growth



## Summary

- Mechanical Reliability of Low-k Materials
  - what is the problem?
  - fracture properties and scaling with density/dielectric constant
- UV Curing Effects on Mechanical Properties
  - fracture path and effects on cohesive and adhesive properties
  - balancing film stress, modulus and fracture properties
- Non-Uniform UV Curing Phenomena
  - evidence for non-uniform curing
  - likely mechanism associated with UV light interference
- CMP Aqueous Chemistry Effects
  - accelerated cracking in aqueous CMP slurries and cleaning solutions
- Die Seal and Crack Stop Structures
  - fracture mechanics approach to preventing dicing damage
- Summary