

"Atomistic modeling of Solid Phase Epitaxial Regrowth using Lattice Kinetic Monte Carlo: Facet formation and strain dependencies

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Outline

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- An atomistic solution: LKMC
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- A look at time evolution
- Fin FETs
- SPER on rectangular shapes.
 - Why corners are different?
 - Introducing strain
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Introduction

- Typicall ultra-shallow junctions produce substrate amorphization.
- Subsequent recrystallization, or Solid Phase Epitaxial Regrowth (SPER) can significantly modified the implanted profiles: SPER simulation is critical.
- SPER depends on the substrate orientation, with ratios 20:10:1 for (100), (110) and (111).



An atomistic solution: LKMC

- Existing non Lattice KMC models accurately predict the one-dimensional SPER.
- Unfortunately, such models do not have different speeds for different orientations, much less faceting.
- In this work, a Lattice KMC model is introduced to overcome these limitations.



The model (I)

- We introduce the silicon lattice in the amorphous/crystalline interface
- A flag "A"morphous or "C"rystalline is set for every of these lattice atoms.
- The transition of "A" atoms to "C" atoms is simulated following an Arrhenius rate
- We follow Drosd and Washburn ideas to define such a rate:
 - Activation energy is constant
 - Prefactor depends on the atom neighborhood.

Drosd and Washburn, J. Appl. Phys. 53, 39 (1982)



The model (II)

- SPER rate is:
- $\upsilon = K(n) \times \exp(-(E + \lambda |\varepsilon_{xy}|) / K_B T)$
 - K(1), prefactor for atoms with two undistorted bonds.
 - K(2), prefactor for atoms needing another one to join in a cluster
 - K(3), prefactor for atoms needing two more to join in a cluster
 - Model implemented in Sentaurus Process KMC



Idea: Two undistorted bonds needed



Planar growth for different orientations (I)



Lines, our work: Appl. Phys. Lett. 95, 123123 (2009) Symbols: G.L. Olson and J. A. Roth. Mater. Sci. Rep. 3 1 (1988)



Planar growth for different orientations (II)



- E = 2.7 eV
- $K(1) = 9.0 \times 10^{16}$ atoms/s
- K(2) = 2.3x10¹⁵ atoms/s
- K(3) = 1.1x10¹¹ atoms/s
- K(1), K(2), K(3) related with the microscopical growth of (100), (110) and (111)

Experimental results from Csepregi et al. J. Appl. Phys. 49, 3906 (1978)



A look at time evolution.





A look at time evolution.





A look at time evolution. -





A look at time evolution.





A look at time evolution. -





A look at time evolution.





A look at time evolution. -





A look at time evolution.





A closer look to the amorphous/crystalline interface





SPER and fins: Faceted growth after fin amorphization.





SPER and fins. Experiment vs. simulation comparison.

Experimental results from Duffy et al. Appl. Phys. Lett. 90, 241912 (2007) Lines: Experiment, symbols, LKMC simulation.





Recrystallization of rectangular shapes:



K.L. Saenger et al. J. Appl. Phys. 101, 084912 (2007)



Why corners are different? Stress/Strain as a reason: lattice distortion





Introducing strain dependency

- The quality of the crystalline lattice template degrades when distorted $\upsilon = K(n) \times \exp(-(E + \lambda |\varepsilon_{xy}|) / K_B T)$
- This is modeled by increasing SPER activation energy by λ|ε_{xy}|
- The absolute value is issued because the lattice is distorted regardless of clockwise or counterclockwise shear strain.



Rectangular shape results: λ =5.



I Martin-Bragado and V. Moroz. Appl. Phys. Lett. 95, 123123 (2009)



The strain at the corner generates the trench





Conclusions

- A LKMC model, based on existing qualitative atomistic ideas, quantitatively explains and reproduces different planar SPER velocities
- The model also satisfactory reproduces facet formation in very thin amorphized fins.
- Finally, it explains anomalous regrowth patters and facet formation in rectangular amorphized (001) and (011) Si substrates.
- Model included in sprocess KMC, 2010.03 release.



Questions?



